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High Performance Molybdenum Disulfide Amorphous Silicon Heterojunction Photodetector

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One important use of layered semiconductors such as molybdenum disulfide (MoS_2) could be in making novel heterojunction devices leading to functionalities unachievable using conventional semiconductors. Here we demonstrate a metal-semiconductor-metal heterojunction photodetector, made of MoS_2 and amorphous silicon (a-Si), with rise and fall times of about 0.3 ms. The transient response does not show persistent (residual) photoconductivity, unlike conventional a-Si devices where it may last 3–5 ms, thus making this heterojunction roughly 10X faster. A photoresponsivity of 210 mA/W is measured at green light, the wavelength used in commercial imaging systems, which is 2–4X larger than that of a-Si and best reported MoS_2 devices. The device could find applications in large area electronics, such as biomedical imaging, where a fast response is critical.

Molybdenum disulfide (MoS_2), a member of the family of layered transition metal dichalcogenides, has long been used as lubricants¹. The interest in electronic and optoelectronic applications of MoS_2 thin films has been kindled by recent demonstrations of high mobility transistors with MoS_2 active layer^{2–3}. While mono-layer MoS_2 is targeted for high end applications such as alternative to silicon transistors^{2–8}, thin film MoS_2 could be suitable for other applications such as displays and thin film sensors⁹. Recently, thin film MoS_2 transistors with mobilities of 70–100 cm^2/Vs were demonstrated^{10,11}, which could be used for driving pixels on displays. Phototransistors with thin film MoS_2 also showed promise for thin film photodetectors, even though their responsivity at the wavelength of 550 nm was about 50 mA/W¹⁰. Among the commercially available thin film photodetectors, amorphous silicon (a-Si) is widely used as the sensing element for several applications including indirect x-ray imagers for radiology^{12–15}. However, persistent photoconductivity, due to structural defects in amorphous silicon, poses several problems such as slow operation speed (tens of frames per second) and image retention (lag), making it challenging to address high speed applications such as fluoroscopy and tomography^{12–15}. Therefore, a fast photodetector would significantly advance the state of the art and expand the range of imaging applications where it could be used. We demonstrate that thin film MoS_2 and amorphous silicon form a heterojunction diode that results in such a high speed photodetector.

Results

We investigated a lateral metal-semiconductor-metal (MSM) device structure consisting of thin film MoS_2 covered with thin film a-Si. This heterostructure made across a Van-der-Waals interface is expected to be robust due to structural stability and chemical inertness of the 2D MoS_2 surface and therefore provide a diode-like device without the need for epitaxy. A schematic of the device structure is shown in Fig. 1a. The energy band diagram is shown in Fig. 1b, where an electron affinity of 3.8 and 4.3 and a bandgap of 1.6 and 1.3 eV for a-Si and MoS_2 , respectively, was assumed^{11,16,17}. In this structure, if photons are absorbed in the a-Si layer, some of photogenerated electrons may diffuse to the a-Si/ MoS_2 junction and subsequently get swept into the MoS_2 side. In general, both layers may contribute to photocurrent. It is important to note, however, that the electrons that get transferred to MoS_2 should move with a higher velocity due to a larger mobility of MoS_2 as compared to a-Si. Bulk mobility of a-Si is about 10 cm^2/Vs ^{18,19}, whereas measured mobility of our MoS_2 thin films were 2–3X higher (see discussions later). The fast moving electrons in MoS_2 are expected to lead to a much faster photoresponse for the a-Si/ MoS_2 photodetector as compared to the ones based on a-Si only.

To test this hypothesis, two MSM devices were fabricated on SiO_2 substrate. The semiconductor layer consists of a 60 nm mechanically exfoliated MoS_2 flake covered by 100 nm a-Si in one device, and the 100 nm a-Si film alone for the second one. The schematic of device cross sections and top view photomicrographs of the fabricated

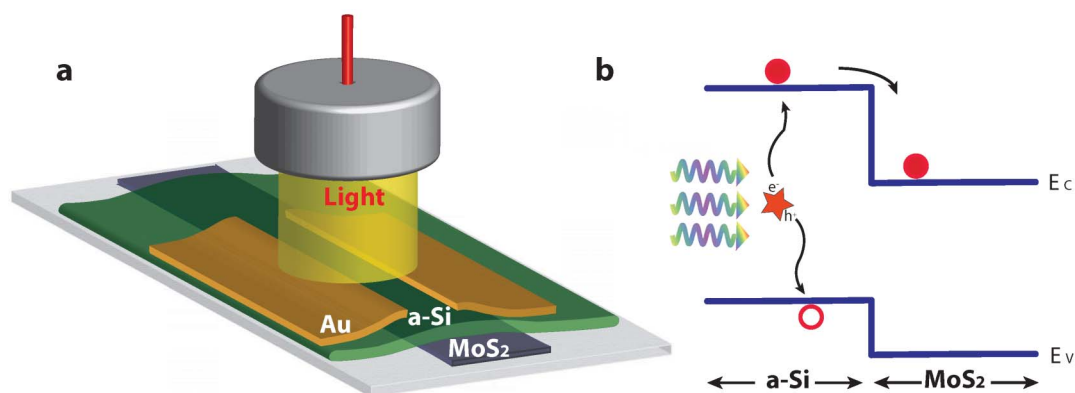


Figure 1 | Operation concept of molybdenum disulfide amorphous silicon heterojunction photodetector. (a) schematic and (b) energy band diagram of a-Si/MoS₂ heterojunction MSM photodetector. The light is incident from a-Si side, optical absorption occurs in a-Si, and photogenerated electrons diffuse to the underlying MoS₂ layer and are transferred across the MoS₂ layer toward a metal contact.

devices are shown in Fig. 2a and b. The length and width of the fabricated devices are roughly 5 and 6 μm , respectively. The current-voltage (IV) characteristics of the devices were measured under dark and illuminated conditions. Figure 2c shows the dark IV characteristics. The dark current of the device with just a-Si active layer is below the noise level of our characterization system which is about 50 fA on average. Low dark currents are expected for a-Si devices as

the resistivity of a-Si is above $10^{10} \Omega\text{cm}^{19}$. The dark current of the hybrid a-Si/MoS₂ device is also below the noise level for voltages less than 0.25 V, and increases over the noise level beyond 0.25 V. It is about 0.3 pA at 1 V. Notably, we also evaluated MSM devices with just a thin film MoS₂ layer, where we found that the dark current is too high (~ 40 pA at 1 V) for it to be particularly useful, see Supplementary Materials and Figure S3. The measured photo IV

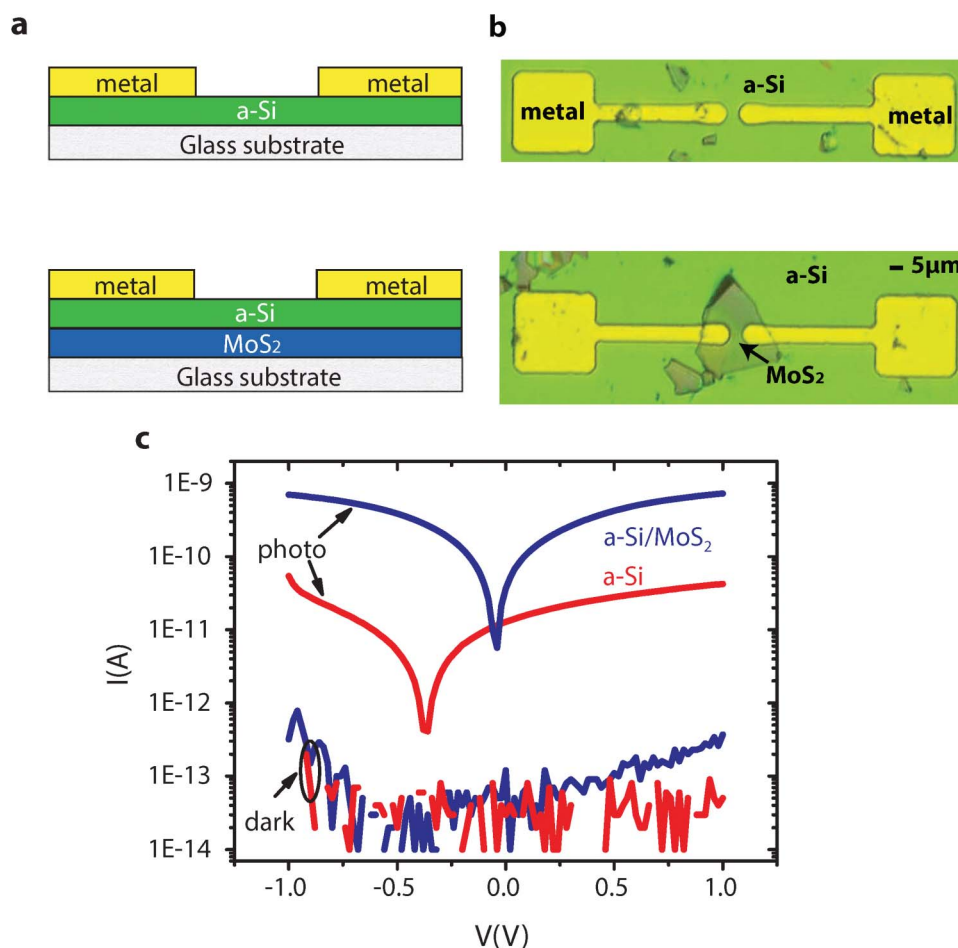


Figure 2 | (a) Schematic of cross sections of two metal-semiconductor-metal (MSM) photodetectors and (b) corresponding top view photomicrograph of devices. The length and width of devices is about 5 and 6 μm , respectively. (c) Measured dark and photo current-voltage (IV) characteristics. The semiconductor layer is 100 nm a-Si film (labeled as a-Si), and 60 nm MoS₂ flake covered by 100 nm a-Si (labeled as a-Si/MoS₂), see Supplementary Materials and Figure S1 for further details.



characteristics of the hybrid a-Si/MoS₂ and a-Si devices are also shown in Fig. 2c. For this measurement, a fiber-coupled broadband light out of a halogen lamp was used. The optical power density was 10 mW/cm². As seen, the photocurrent of the hybrid a-Si/MoS₂ device is about one order of magnitude larger than that of the a-Si device.

To obtain further insight into the device operation, we measured photoresponse of devices at three wavelengths corresponding to blue, green, and red colors. For these measurements, we used standard LEDs as light sources and the incident power was 0.4 mW/cm². Photoresponsivity, calculated as the ratio of photocurrent to incident power, is shown in Fig. 3a. As seen, the responsivity of the hybrid a-Si/MoS₂ device is larger than that of the a-Si device for all wavelengths. For example, at $\lambda = 400$ nm, they are about 150 and 60 mA/W for the hybrid a-Si/MoS₂ and the a-Si device, respectively. For $\lambda = 550$ nm, the responsivity is maximum which is about 210 mA/W. This responsivity is 4× larger than that of best MoS₂ devices¹⁰, and 2× larger than that of a-Si devices reported here. The green wavelength is of special interest as x-ray imagers operate at this frequency. At the wavelength of 630 nm, the photoresponse of the hybrid device is seven times larger than that of the a-Si device. The reason for this larger gain is that the a-Si layer does only absorb a part of the incident light at 630 nm and thus the underlying MoS₂ is also contributing to photogeneration. This is confirmed by transmittance measurements as a function of wavelength as shown in Fig. 4. We also investigated the photoresponse of the a-Si/MoS₂ device, with 100 nm a-Si, when the incident power changes. Fig. 3b shows the photocurrent as a function of incident power from 50 to 400 μ W/cm². As seen, it is fairly linear versus the input power.

Next we measured the transient response of both set of devices by pulsing the incident light by biasing the LEDs with a 3 V, 100 Hz square voltage out of a function generator. The results are shown in Figs. 3c and 3d. As seen, the response of the a-Si/MoS₂ device does not show residual conductivity, i.e. the pulses are well defined and on/off levels are more stable and cleaner. However, there is persistent (residual) conductivity in the response of the a-Si sample. Figure 3e shows a close view of two pulses. It is evident that the residual conductivity of the a-Si device may last for about 3 msec. Indeed, the existence of residual conductivity in a-Si photodetectors of the order of a few msec is very well-known^{12–15}. For example, in Ref. [14], a value of 5 msec was reported. To determine the time of residual conductivity in our devices, we plotted the statistical distribution of the residual conductivity of the a-Si device, shown in Figure 3f, which clearly shows a 3–5 msec average. Note that, for some pulses, the conductivity never reaches the steady state before the next pulse arrives. We have ignored those pulses. Thus this estimated average time should be considered to be a lower bound. For practical imaging applications the residual conductivity of a-Si limits the operation speed, because one has to wait 3–5 msec for it to vanish.

Now, let us look at the rise and fall times. We defined the fall and rise times as the time it takes to switch from one level to about 90% of the second level. Figure 3g shows one example, where the response pulses are normalized. As seen, a response time of 0.3 ms is extracted for both devices. Using the data from Figs 3c and 3d, we also obtained statistical information of the transient response. The data are summarized in Supplementary Figures S5–6. In addition, the transient response of the hybrid device can be modeled with an exponential function, $y = y_0 + A \exp(-t/t_0)$, where t_0 is the time constant. The t_0 is about 170 μ sec and its statistical distribution is shown in Supplementary Fig. S7. These observations can now be summarized in the following way: both photodetectors show similar rise and fall times which are of the order of 0.3 msec. However for a-Si devices, there exists a residual photoconductivity, as it is well-known in literature^{12–15}, that limits its speed to 3–5 msec. For the heterojunction device, on the other hand, the residual conductivity is not present,

making it possible to use it right after the initial fall time. Comparing these two timings, we conclude that the imaging speed can be increased by 10X.

Thus far, based on the photoresponsivity data, one may conclude that the junction between the MoS₂ and a-Si enhances the performance by ensuring that a part of photogenerated electrons inside the a-Si that survive and diffuse to the junction are swept to the MoS₂ side. To test what happens for a thicker a-Si, we made another set of devices with 300 nm a-Si with the same dimensions as those of the first set (Note that the diffusion length of a-Si is about 300 nm¹⁹). Supplementary Fig. S4 shows the photoresponsivity data measured under similar conditions like before. As seen, devices with 300 nm a-Si, with and without MoS₂, give the same photoresponse for all three incident lights. Therefore, in this case, photogenerated electrons completely reside inside the a-Si layer. Compared to the devices with 100 nm a-Si, the ones with 300 nm a-Si give larger photocurrents and, for example, at $\lambda = 550$ nm the responsivity is about 1 A/W. As for the dynamic response, both devices show similar speed of about 3 msec, see Supplementary Figures S8 and S9. This observation indicates that the MoS₂ junction is beneficial and results in a fast photoresponse when the thickness of a-Si is less than the diffusion length of electrons, so that electrons survive to reach the junction and be transferred to the MoS₂. Similar transient responses were measured for all three wavelengths; see Supplementary Figures S8–9. Notably, when a-Si thickness is 300 nm, the photocurrent is higher compared to when it is 100 nm because the volume of absorption is now much larger. Therefore, we conclude that there exists a trade-off between speed and amplitude of photocurrent in this heterojunction. While the larger mobility of MoS₂ helps increase the speed of the device, a smaller thickness of a-Si, which is needed to ensure that electrons can reach the junction before recombining, could lead to reduced absorption and therefore smaller photocurrent. By varying the thickness of a-Si and MoS₂ layers, it should be possible to obtain an optimized scenario where a substantial increase in speed can be gained without giving up the photocurrent significantly.

Before concluding, we provide additional experimental measurements that elucidate the physics of charge transfer from a-Si into the underlying MoS₂ at the junction. Fig. 4a shows two sets of devices with 100 and 300 nm a-Si, with and without MoS₂, that were subjected to an incident light of $\lambda = 400$ nm. The reason to choose this wavelength is that it is completely absorbed in the top a-Si layer without reaching the MoS₂. Fig. 4b shows transmission measurements of a-Si in the UV-visible range where below 420 nm the transmission is virtually zero. This implies that in the a-Si/MoS₂ devices the MoS₂ is not receiving any blue light. However, MoS₂ is electrically active as deduced from the IV data shown in Fig. 4c, for the devices with 100 nm a-Si. As seen, the IV of the a-Si device is linear, but that of the hybrid a-Si/MoS₂ device is non-linear and larger. This validates the assumption made in the beginning that MoS₂ and a-Si form a diode where part of photogenerated electrons diffuse from a-Si towards the junction and are swept into the MoS₂. When the a-Si is 300 nm thick, both IVs are the same and fairly linear, see Fig. 4d. This implies that there is no charge transfer from 300 nm a-Si into the MoS₂ layer. In addition, we mentioned that transferred electrons move with a higher mobility in MoS₂ and this contributes to the larger photocurrent of the hybrid device. To support this argument, we made a bottom-gate transistor out of thin film MoS₂ on thermally grown SiO₂ as the gate dielectric. The schematic of cross section and the top view photomicrograph of the transistor are shown in Fig. 4e. The transfer characteristic is shown in Fig. 4f. A mobility of 28.6 cm²/Vs was extracted from the linear plot of the curve, as shown in the inset of Fig. 4f. This value is larger than that of the bulk of a-Si, about 10 cm²/Vs^{18,19}. Thus the higher mobility of MoS₂ is effective in enhancing the photoresponse of the hybrid photodetector.

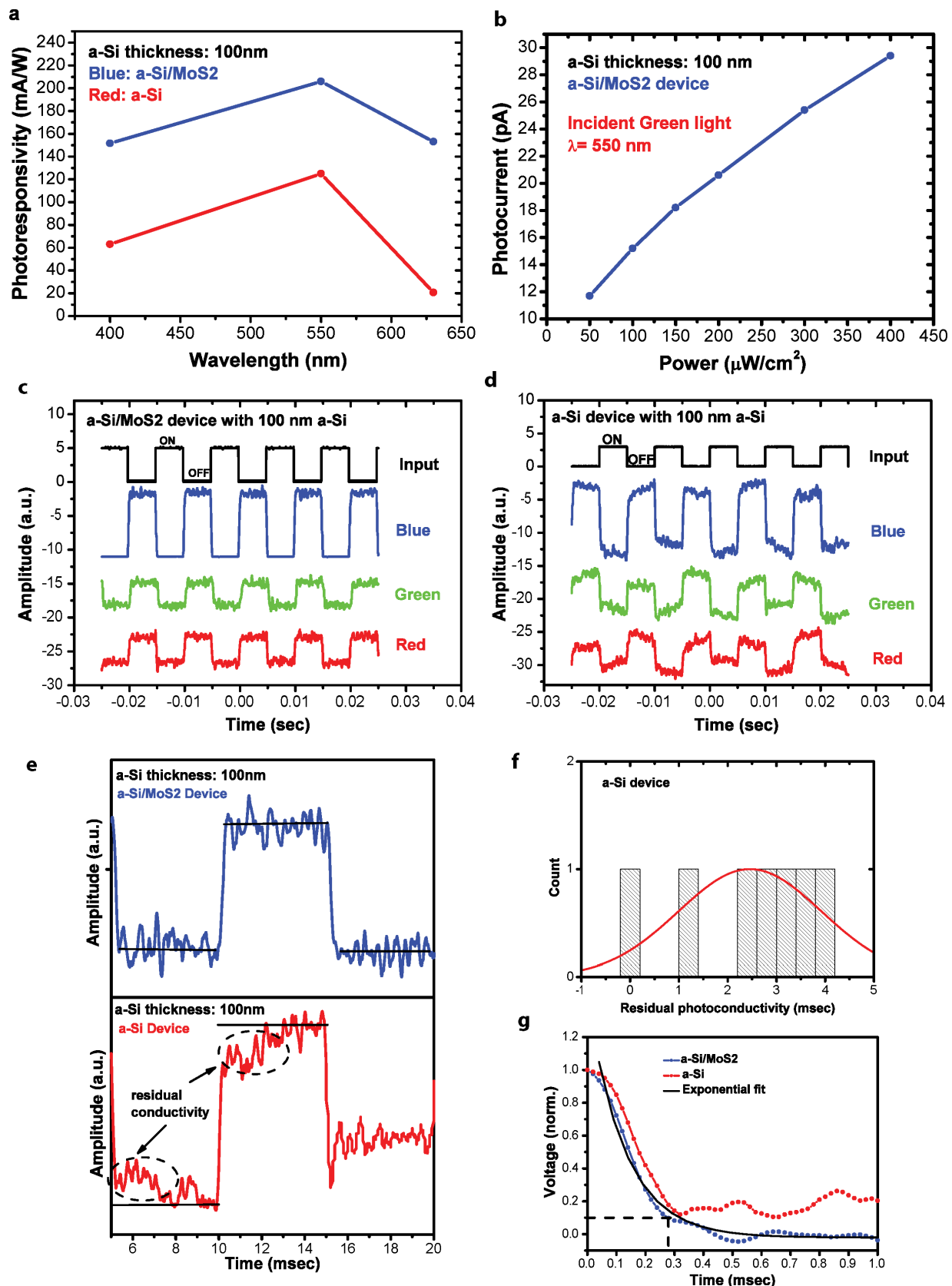


Figure 3 | (a) Measured photoresponsivity of a-Si/MoS₂ and a-Si MSM devices for three wavelengths corresponding to blue, green, and red colors; the thickness of a-Si is 100 nm. Photocurrents at the applied voltage of 1 V were taken for calculations. The incident power was 0.4 mW/cm². (b) photocurrent of the a-Si/MoS₂ device as a function of incident power. Applied voltage was 1 V. (c) and (d) transient responses of the two devices for three incident wavelengths. For part (d), the voltage on the a-Si device was increased to obtain photocurrent pulse amplitudes comparable to that of the hybrid device. (e) The close view of two pulses highlighting the residual conductivity of the a-Si device. (f) statistical distribution of the residual conductivity of the a-Si device. (g) response time extraction and fitting with an exponential function. In parts (a) and (b), lines are for the eye guide only.

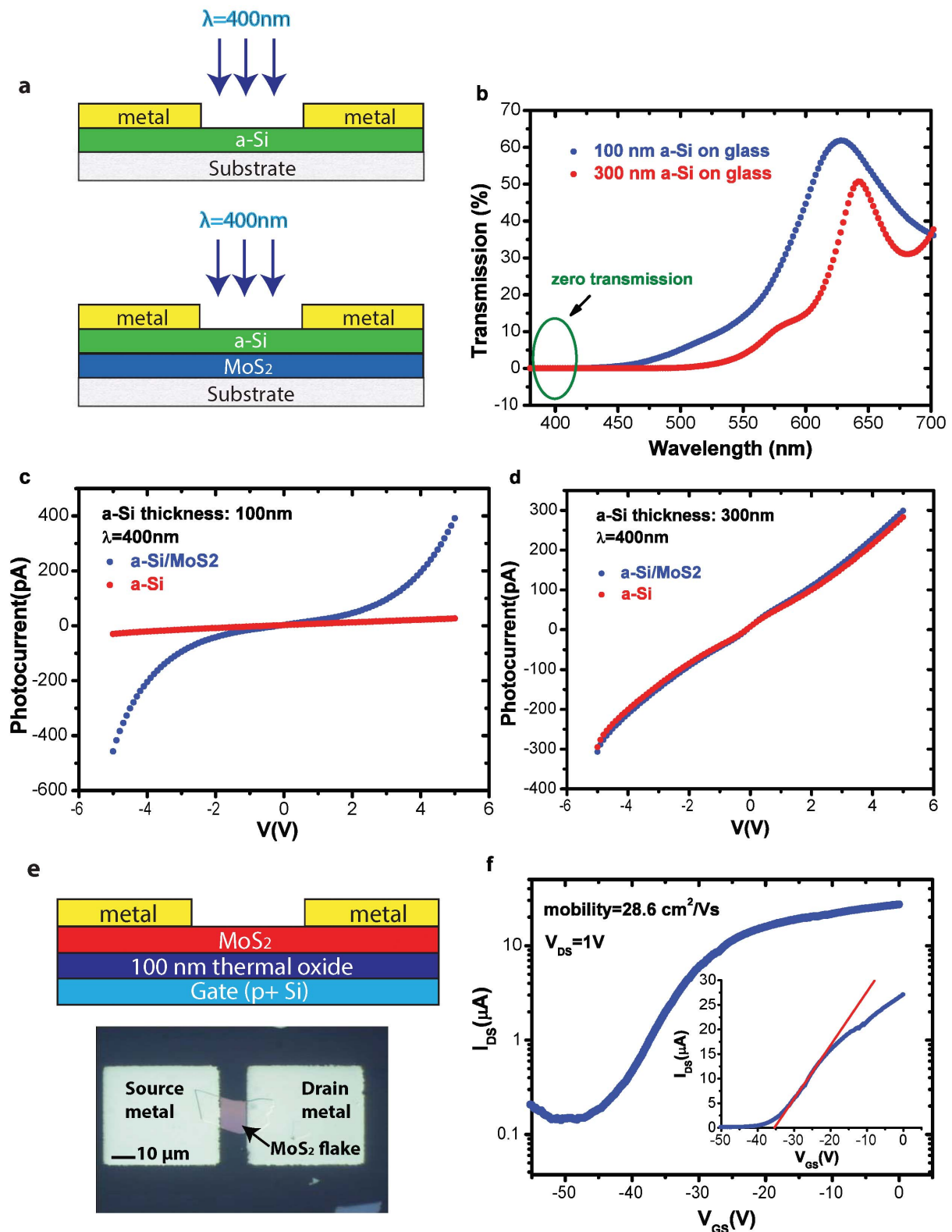


Figure 4 | Experimental data that support the proposed concept of operation of the photodetector. (a) Schematic diagram of devices subjected to blue incident light. (b) Measured UV-visible transmission of 100 and 300 nm a-Si thin films, used in the two sets of devices, with and without MoS₂ flakes. (c) Measured photo IV of devices with 100 nm a-Si, showing that the IV of the a-Si/MoS₂ device is non-linear and MoS₂ is effective in boosting the photocurrent. (d) Measured photo IV of devices with 300 nm a-Si, showing that MoS₂ is not effective in enhancing the photocurrent. (e) schematic cross section and top view photomicrograph of the bottom-gate transistor, made of a 73 nm thick MoS₂ flake. (f) Measured transfer characteristic where a mobility of $28.6\text{ cm}^2/\text{Vs}$ was extracted.



Discussion

The a-Si/MoS₂ photodetector could be used for biomedical imaging applications where a fast photoresponse is required. For example, currently, flat-panel x-ray imagers based on a-Si p-i-n photodetectors operate at frame rates in the range of 10–100 Hz, limited by the slow photoresponse of a-Si. The detector presented here may offer a speed of operation up to several kHz, considering its rise and fall times of 0.3 ms. The fast response also allows shorter x-ray exposure times to patients which helps to reduce the health hazards of x-ray radiation. From technology point of view, the advantage of this detector is its simplicity of fabrication which can be easily integrated with other components and readout circuits. From device and structure point of view, the layered material MoS₂ and, in general, the family of metal dichalcogenides, do not form strong chemical bonds with other materials. The bonding is mainly due to Van-der-Waals forces. Thus they may enable a new class of devices, Van-der-Waals heterojunctions, by forming a wide range of 2D/2D and 2D/3D heterostructures²⁰. Finally, for practical applications, large area growth methods such as CVD should be developed. Research in this direction is underway by several groups.

In conclusion, we demonstrated that molybdenum disulfide forms a heterojunction diode with amorphous silicon which leads to a 10X faster response compared to the state of the art. The device has a photoresponsivity of about 210 mA/W at the wavelength of 550 nm. In addition, it has a low dark current of 0.3 pA at 1 V bias. Going forward, such devices could be integrated with MoS₂ transistors, as readout circuits and pixel amplifiers, leading to a monolithic large area technology based on layered semiconductors.

Methods

Flakes of MoS₂ were exfoliated on SiO₂ substrates from a piece of MoS₂ crystal (provided by the SPI Supplies, www.2spi.com) using the scotch-tape mechanical cleavage method. Subsequently, the thin film MoS₂ flakes were coated with amorphous silicon (a-Si) thin films, deposited by plasma-enhanced chemical vapour deposition (PECVD) using silane gas source at 260 °C. Metal contacts were formed by the conventional lift-off technique, where Ti (20 nm) and Au (160 nm) were deposited by electron-beam evaporation at room temperature. Current-voltage characteristics of devices were measured by an Agilent 4155C semiconductor parameter analyzer, under dark and illuminated conditions. For the latter, a fiber-coupled broadband light out of a halogen lamp was used. The optical power density was about 10 mW/cm². We also measured photoresponse of devices at three wavelengths corresponding to blue, green, and red colors. For these measurements, we used standard LEDs as light sources and the incident power was about 0.4 mW/cm². All measurements were done at room temperature.

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Author contributions

M.E. and S.S. devised the device concept and initiated the research. M.E. worked on device fabrication, performed measurements, and wrote the manuscript. All the authors read and commented on the manuscript.

Additional information

Supplementary information accompanies this paper at <http://www.nature.com/scientificreports>

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